R_0A product for PbS and PbSe abrupt p-n junctions

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The effect is analysed of the concentration departs on the R_0A product of PbS and PbSe abrupt p-n junctions. Calculations are performed for temperatures of 77, 200, and 300 K. The influences of the diffusion current (for radiative and Auger recombination), of the tunneling and depletion layer currents are considered.

1. Introduction

The investigations of the lead chalcogenide compounds started at date back to the early history of the semiconductor research. The first paper published on the subject appeared in 1874, when Braun reported asymmetrical nature of conduction between metal electrodes and the natural crystal of galena (lead sulphide) [1]. The wide application of lead sulphide in the early days of radio is well known. During World War II in Germany, next in the United States (Northwestern University) and in England (Admiralty Research Laboratory) the PbS, PbSe, and PbTe photoconductive infrared detectors having the polycrystalline thin film form were produced for the spectral range of 1.7 to 7 μ m. These detectors have especially found military applications. Extensive reviews of the properties and applications of lead chalcogenides detectors are given in works [2, 3]*.

The photovoltaic detectors appeared to have a higher detectivity than photoconductive detectors within lower temperature range. The theoretical limited parameters of PbTe photovoltaic detectors were determined in papers [4, 5]. Also PbS and PbSe are used for production of photovoltaic detectors [6–8], the theoretical limiting parameters of these detectors are, however, not available.

In the present work the effect of the doping concentration on the zero bias junction resistance area product R_0A is analyzed for abrupt p-n junctions in PbS and PbSe.

2. R_0A product for abrupt p-n junction

The total dark current density flowing through the junction is given by

$$J = J_D + J_{GR} + J_T + J_L, (1)$$

^{*} In a recent paper ZIEP O. and MOCKER M., (Phys. Stat. Sol. (b) 98 (1980), 33) elaborated a modified theory of Anger recombination in lead chalcogenides.

where the particular components denote diffusion, generation-recombination in the depletion layer, tunneling and leakage current density, respectively. The latter may be due to bulk as well as to surface defects of the material. The value of J_L cannot to calculated theoretically, but when a suitable diode technology and construction are used, the contribution of the J_L component is negligible.

The R_0A product determined by the diffusion current in the case of radiative recombination is [9]

$$(R_0 A)_R = \frac{(kT)^{1/2}}{q^{3/2} n_i^2 \mu^{1/2} R^{1/2}} \sqrt{n_{\text{ef}}},$$
 (2)

where $n_{\rm ef} = np/(n+2\sqrt{np}+p)$ is the effective doping concentration (n_i and p are the electron and hole majority carrier concentrations on both sides of the junction), q is the electron charge, μ is the average value of the mobility of electrons and holes, k is Boltzmann's constant, T is the temperature, n_i is the intrinsic carrier concentration and R is the coefficient of radiative recombination.

In the case of equal electron and hole masses $m_e^* \approx m_h^*$ the $R = R_n = R_p$ coefficient may be written in the form [10]

$$R = \frac{1 \times 10^{15} \, \overline{n} E_g^2}{(kT)^{3/2} K^{1/2} (2 + 1/K)^{3/2} (m^*/m_0)^{5/2}}, \tag{3}$$

where \overline{n} is the refractive index, E_g is the energy gap, $K = m_1^*/m_i^*$ is the effective mass anisotropy coefficient and m_0 is the mass of free electron. The m^* can be determined if the longitudinal m_1^* and transveral m_i^* components of the effective mass are known, since $m^* = [1/3(2/m_i^* + 1/m_1^*)^{-1}]$. In eq. (3) the values of kT and E_g should be expressed in electronovolts.

The intrinsic carries concentration is given by [11]

$$n_i = 2\left(\frac{2\pi kT}{h^2}\right)^{3/2} (m_{dn}^* m_{dp}^*)^{3/4} \exp\left[-\frac{E_g}{2kT}\right],$$
 (4)

where m_{dn}^* and m_{dp}^* denote the density-of-states effective masses of electrons and holes, respectively. The formula (4) is valid for $E_g \gg kT$, being fulfilled in the case of lead chalcogenides. The values of density-of-states effective masses can be obtained from $m_d^* = N^{2/3} (m_l^* m_l^{*2})^{1/3}$, where N=4 is the number of equivalent band extrema.

The R_0A product determined by the diffusion current in the case of Auger recombination is [9]

$$(R_0 A)_A = \frac{(kT)^{1/2}}{2q^{3/2} n_i^2 \mu^{1/2} C^{1/2}},$$
 (5)

where C is the Auger recombination coefficient.

Due to the mirror symmetry of the conduction and valence bands $C = C_n$

 $= C_p$. In the case of onevalley collision recombination ($K \approx 1$) [12]

$$C = \frac{1}{An_i^2} \left(\frac{kT}{E_g}\right)^{3/2} \exp\left[-\frac{3E_g}{2kT}\right],\tag{6}$$

where $A = (1; 5) \times 10^{-13} s$ for PbS and PbSe, respectively

The R_0A product for abrupt p-n junctions determined by depletion layer is given by the formula [13]

$$(R_0 A)_{GR} = \frac{E_g^{1/2} \tau_0}{q \dot{n}_i (2\varepsilon_s)^{1/2}} \sqrt{n'_{\text{ef}}}, \tag{7}$$

where ε_s denotes the static dielectric constant and $n'_{\rm ef} = np/(n+p)$ is the effective concentration. In the formula (7) τ_0 , i.e. the time determining the recombination by the Shockley-Read centres, is a parameter difficult to determine.

The R_0A product determined by tunneling is given by [9]

$$(R_0 A)_T = \frac{h^2 \varepsilon_s^{1/2}}{2m_{\text{tu}}^{*1/2} q^3 \sqrt{n_{\text{ef}}'}} \exp\left[\frac{8\pi m_{\text{tu}}^{1/2} E_g \varepsilon_s^{1/2}}{3qh \sqrt{n_{\text{ef}}'}}\right], \tag{8}$$

where h is Planck's constant and $m_{\rm tu}^* \approx m^*/2$ is the tunneling effective mass.

3. Numerical results of calculations and discussion

The dependence of the R_0A product on the effective concentration of dopants at 77, 200, and 300 K was calculated using the formulae (2), (5), (7), and (8); the parameters are listed in table. The effective masses were determined basing

Table. Material parameters of PbS and PbSe

	T [K]	$egin{aligned} E_g\ [\mathrm{eV}] \end{aligned}$	m_1^*/m_0	m_t^*/m_0	K	$\mu \ [\mathrm{m^2/Vs}]$	\overline{n}	ϵ_{s}	ϵ_{∞}
PbS	77	0.315	0.114	0.084	1.35	1.3	4.5	180	18.4
	200	0.373	0.135	0.099	1.35	0.15	4.4	180	17.6
	300	0.422	0.153	0.112	1.35	0.06	4.3	180	17.2
PbSe	77	0.175	0.081	0.043	1.9	1.5	5.2	250	25.2
	200	0.233	0.107	0.057	1.9	0.27	5.1	250	23.8
	300	0.282	0.130	0.070	1.9	0.10	5.0	250	22.8

on paper [14]. The effective masses of electrons and holes were expressed in their mean values; it has been assumed that the effective masses vary proportionally to the width of the energy gap. This approximation is justified by the results of the theoretical Kane model and experimental studies of the band structure of lead chalcogenide compounds [15]. The values of the energy gap E_g , effective mass anisotropy coefficients K, carrier mobilities μ , refractive

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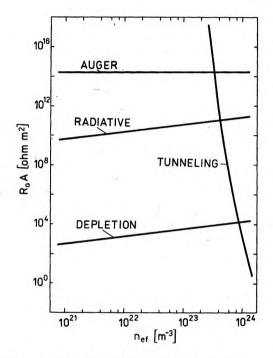
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index \overline{n} , static dielectric constant ε_s , and of high frequency dielectric constant ε_{∞} were assumed after [15, 16].

It is expected that τ_0 (see eq. (7)) depends on the temperature as well as the defect concentration. This dependence is not known for lead chalcogenides. In the calculations of $(R_0A)_{GR}$ it was assumed that $\tau_0 = 10^{-8}$ s. The same order of time is often assumed for lead chalcogenides [13, 17].

The results of calculations are presented graphically in figs. 1-4 for 77,



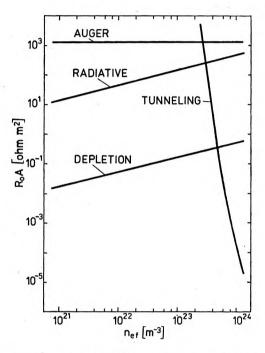


Fig. 1. The dependence of the zero bias resistance area product, R_0A , on the effective doping concentrations for abrupt PbS junctions at 77 K

Fig. 2. The dependence of the zero bias resistance area product, R_0A , on the effective doping concentrations for abrupt PbSe junctions at 77 K. The experimental value is taken from paper [19] (\bigcirc)

200, and 300 K. At 77 K the R_0A product for PbS and PbSe abrupt p-n junctions is determined by generation current of the depletion layer. For the radiative and Auger recombinations the theoretical estimates yield a few orders of magnitude larger values of the R_0A product. The tunneling current produces an abrupt lowering of R_0A at the effective doping concentrations of 8×10^{23} m⁻³ and 4×10^{23} m⁻³ for PbS and PbSe p-n junctions, respectively. To obtain possibly high values of the zero bias resistance of the junctions the technological process of photovoltaic detectors preparation should be connected so that the concentrations of dopants be slightly below those concentrations. At such concentrations the influence of the Burstein-Moss shift is possible. The Fermi

level achieves the band edges at carrier concentrations of $(3.3, 1.4) \times 10^{23}$ m⁻³ at 77 K, $(1.9, 1.0) \times 10^{24}$ m⁻³ at 200 K and $(4.3, 2.7) \times 10^{24}$ m⁻³ at 300 K for PbS and PbSe, respectively [18].

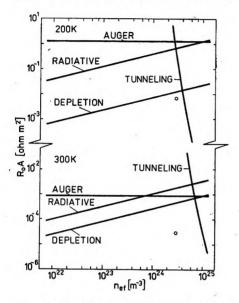


Fig. 3. The dependence of the zero bias resistance area product, R_0A , on the effective doping concentrations for abrupt PbS junctions at 200 and 300 K. The experimental values are taken from paper [6] (\bigcirc)

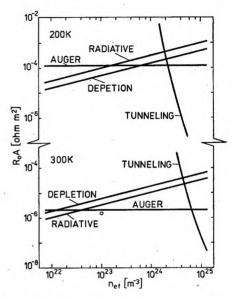


Fig. 4. The dependence of the zero bias resistance area product, R_0A , on the effective doping concentrations for abrupt PbSe junctions at 200 and 300 K. The experimental values are taken from paper [19] (\bigcirc)

At 200 and 300 K the R_0A product for PbS junctions is determined, as before, by generation-recombination current of the depletion layer.

At 200 K the effect of the depletion layer for PbSe junction is significant for $n_{\rm ef} < 6 \times 10^{23} \ {\rm m}^{-3}$. In the range of higher dopant concentrations the effect of Auger recombination on the R_0A product is revealed. At room temperature the Auger process is decisive for the value of the R_0A product in a wide range of concentrations.

From figs. 1-4 it is apparent that the optimum effective doping concentrations (for which the R_0A product takes the maximum value) increases with the temperature.

4. Conclusions

From the analysis of the effect of doping concentrations at both sides of the abrupt p-n junction in PbS and PbSe on the value of the R_0A product it follows that at liquid nitrogen temperature the generation-recombination current within the junction depletion layer plays a dominant role. A further increase of the R_0A product can be achieved by increasing the time τ_0 see (eq. (7)), which, in turn, is limited by the technology process used in junction fabri-

cation. The diffusion processes appears to be important at the range of higher temperatures.

It is difficult to compare results of theoretical calculations with experimental data (known from publications), since in the experimental works the doping concentrations at the both sides of the junction are not given and the junction profile is not always described (the abrupt junctions are considered in this work). Figs. 2-4 show the experimental values from only two papers: for linearly graded PbS junction [7] and for PbSe junction [19]. In the latter the PbSe junction profile was not found [19]. A rough comparison between the calculated and experimental data [6, 19] shows that the construction of higher quality p-n junctions seems to be possible, especially for PbSe at lower temperature.

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Произведение R_0A скачкообразных p-n переходов PbS и PbSe

В работе анализируется влияние уровня концентрации примесей на произведение R_9A скачкообрабаьт p-n переходов, изготовленных в PbS и PbSe. Расчёты произведены для температуры 77, 200 и 300 К. Рассмотрено влияние диффузионного (для излучательной рекомбинации и рекомбинации Оже), туннельного и генерационно-рекомбенационного тока перехода.